

{040/110} Facet Isotype Heterojunctions with Monoclinic Scheelite BiVO_4

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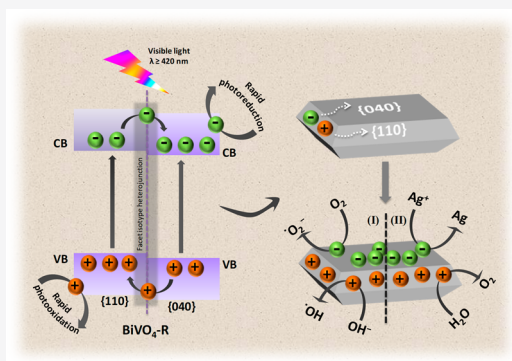
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ABSTRACT: Controlling the phase, crystallinity, and microstructure and fabricating a facet isotype heterojunction with a proscribed reduction–oxidation facet exposure factor have a strong constructive effect toward photoexciton separation and migration. In this respect, here diverse synthetic courses such as calcination ($\text{BiVO}_4\text{-C}$), hydrothermal treatment ($\text{BiVO}_4\text{-H}$), and a reflux method ($\text{BiVO}_4\text{-R}$) are introduced to fabricate various hierarchical morphologies of highly crystalline monoclinic scheelite bismuth vanadate (BiVO_4) with different redox facet exposure factors that have been well established by X-ray diffraction, Fourier transform infrared spectroscopy, field emission scanning electron microscopy, and transmission electron microscopy analysis. The analytical and experimental investigations revealed superior photocatalytic upshots of a $\text{BiVO}_4\text{-R}$ {040/110} facet isotype heterojunction toward levofloxacin (LVF) detoxification (71.2%, 120 min) and the water oxidation reaction (530.6 μmol , 120 min) relative to $\text{BiVO}_4\text{-C}$ (42.3%, 434.2 μmol) and $\text{BiVO}_4\text{-H}$ (60.4%, 494.8 μmol). Accordingly, the $\text{BiVO}_4\text{-R}$ {040/110} facet isotype heterojunction (145.6 $\mu\text{A}/\text{cm}^2$) expressed an enhanced photocurrent in comparison to pristine $\text{BiVO}_4\text{-C}$ (75.5 $\mu\text{A}/\text{cm}^2$) and $\text{BiVO}_4\text{-H}$ (113.1 $\mu\text{A}/\text{cm}^2$). The superior photocatalytic redox efficiency was attributed to well-exposed {040} reduction and {110} oxidation facets and a superior relative {040} facet exposure factor provoking an enhanced charge carrier separation over a $\text{BiVO}_4\text{-R}$ {040/110} facet isotype heterojunction. The spatial exciton separation over the $\text{BiVO}_4\text{-R}$ sample was well established by numerous analytical and experimental investigations. The effectual associations among physicochemical, photoelectrochemical properties, {040/110} facet isotype heterojunction, relative reduction–oxidation facet exposure factor, and photocatalytic performances of fabricated BiVO_4 microstructures were well established, and the upshots of this research were discussed finely. The research signifies an effectual direction for morphology and relative reduction–oxidation facet exposure factor controlled fabrication of facet isotype heterojunction based materials for superior photocatalysis and could be advantageous for supplementary research areas.



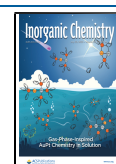
1. INTRODUCTION

Challenging the current dilemma of energy-environmental catalysis, redox course driven semiconductor-based photocatalysis has emerged as an eco-efficient and sustainable resolution.^{1,2} Despite the enormous progress that has been made, surface incompatibility provoked rapid exciton recombination suppresses the expected redox efficiency of traditional heterojunction materials. Moreover, to triumph over the inefficiencies of traditional heterojunctions, n–n isotype heterojunctions between materials with the same charge carrier such as $\text{ZnWO}_4/\text{Bi}_2\text{WO}_6$, $\text{BiPO}_4/\text{Bi}_2\text{WO}_6$, $\text{g-C}_3\text{N}_4/\text{TiO}_2$, and $\text{Bi}_{3.64}\text{Mo}_{0.36}\text{O}_{6.55}/\text{Bi}_2\text{MoO}_6$, etc. and crystal phase isotype heterojunctions between different crystal phases of a pristine material such as $\alpha\text{-Ga}_2\text{O}_3/\beta\text{-Ga}_2\text{O}_3$, $\text{CN-T}/\text{CN-U}$, GCN/CCN , and $\text{TiO}_2(\text{A})/\text{TiO}_2(\text{R})$ have also been introduced.^{3–10} Captivatingly, an $\text{InS}_3/\text{BiVO}_4$ n–n isotype and an $\text{M-BiVO}_4/\text{T-BiVO}_4$ crystal phase isotype heterojunction toward antibiotic detoxification and water oxidation were successfully reported by our group.^{11,12} Superior interfacial compatibilities and a

space charge accumulation/depletion region promoted exciton separation over these heterojunction heighten the photo redox efficiency. Nevertheless, to broaden the research further, a facet isotype heterojunction was introduced over different exposed reduction–oxidation facets of pristine semiconducting materials. Various widely reported materials with crystal facet engineering include TiO_2 , BiOCl , BiVO_4 , NaTaO_3 , ZnFe_2O_4 , CdS , BiOIO_3 , etc.^{13–22} The thermodynamic separation of excitons over the active facets facilitates the overall redox process.

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From this perspective, the monoclinic scheelite BiVO_4 has evolved as an appealing metal oxide photocatalytic material for a crystal facet engineered isotype heterojunction, as its efficiency has been reported to be greatly exaggerated by alteration of the microstructure.^{16,17,23–28} The controlled exposure of oxidation {110} and reduction facets {040} of monoclinic scheelite BiVO_4 adds constructive effects toward exciton separation over the respective energy band edges of an inbuilt facet isotype heterojunction.^{16–18,27–31} Moreover, a narrow band gap of 2.3–2.4 eV, superior photostability, nontoxicity, and superior $\text{Bi}_{6s}-\text{O}_{2p}$ hybrid orbital aggravated visible light absorption ability are the reasonable key factors making the monoclinic scheelite BiVO_4 a desirable alternative to conventional UV-active titania (TiO_2) and tetragonal scheelite/zircon phase BiVO_4 based photocatalysts in the progressive research field.^{1,2,11} Furthermore, Bi–O bond distortion provoked enhanced exciton separation and the destructive effect of $[\text{VO}_4]$ tetrahedron provoked superior water oxidation tendency of the monoclinic scheelite BiVO_4 attest to its novelty.^{11,23}

Nevertheless, research exposing the effective involvement of physicochemical features, electrochemical properties, {040/110} crystal facet isotype heterojunctions, and the relative exposure factors of reduction–oxidation facets of BiVO_4 toward superior photocatalytic levofloxacin detoxification and water oxidation has been unavailable so far. Accounting for the above scenario herein, we fabricated various hierarchical morphologies of highly crystalline monoclinic scheelite BiVO_4 with varying redox facet exposure factors via diverse synthetic routes such as calcination ($\text{BiVO}_4\text{-C}$), hydrothermal treatment ($\text{BiVO}_4\text{-H}$), and a reflux method ($\text{BiVO}_4\text{-R}$) and investigated the efficiencies of the fabricated microstructure toward levofloxacin detoxification and the water oxidation reaction. The crystallographic features contributing toward redox efficiency have been well determined by X-ray diffraction (XRD) analysis. The photophysical properties were well established by a UV–visible diffuse reflectance spectroscopy (UV–vis DRS) analysis. A Mott–Schottky (MS) analysis was executed to investigate the charge carrier characteristics of fabricated samples. The nature of the exciton separation over the photocatalysts was well exposed by photoluminescence (PL), electrochemical impedance spectroscopy (EIS), carrier density, linear sweep voltammetry (LSV), and Bode analysis. The microstructures, surface morphologies, existence of {040/110} facet isotype heterojunctions, and relative facet exposure factors of the fabricated samples were well observed by field emission scanning electron microscopy (FESEM) analysis. Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) analyses were carried out to validate the {040/110} facet isotype heterojunction provoked exciton separation process. A comparative investigation on the effective association among physicochemical and electrochemical properties, {040/110} crystal facet isotype heterojunctions, and relative exposure factors of oxidation–reduction facets toward the superior photocatalytic redox efficiency of BiVO_4 was neatly ascertained and discussed.

2. EXPERIMENTAL SECTION

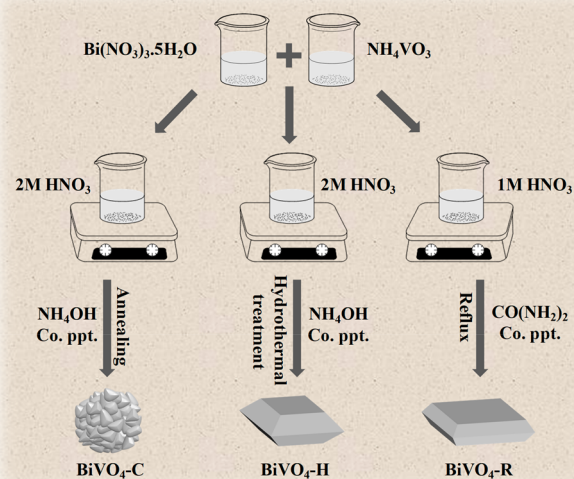
2.1. Chemicals and Reagents. Analytical grades of bismuth nitrate pentahydrate ($\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$, 99%), ammonium metavanadate (NH_4VO_3 , 99%), urea ($\text{CO}(\text{NH}_2)_2$, 99%), ammonium hydroxide (NH_4OH , 28–30%), nitric acid (HNO_3 , 69%), and silver nitrate

(AgNO_3 , 99%) from Merck were used without any additional refinement for material fabrication. Double-distilled deionized water ($\text{DI-H}_2\text{O}$) was used throughout the experiments. Analytical grade ethanol ($\text{C}_2\text{H}_5\text{OH}$, 99.9%) was used for washing.

2.2. Fabrication of Hierarchical Microstructures of BiVO_4 .

About 20 mmol of $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ and 20 mmol of NH_4VO_3 were added to each of three distinct beakers containing 165 mL of 2 M dilute HNO_3 (SOL-A), 160 mL of 2 M dilute HNO_3 (SOL-B), and 106 mL of 1 M dilute HNO_3 (SOL-C). The SOL-A was stirred for 2 h magnetically until a homogeneous yellow solution was obtained followed by slow addition of NH_4OH until pH 9. The precipitate that formed was washed (H_2O , $\text{C}_2\text{H}_5\text{OH}$) and dried in a hot air oven (70 °C, 24 h). The product was then finely ground followed by calcination at 500 °C for 5 h (10 °C/min). The sample was collected and assigned the name $\text{BiVO}_4\text{-C}$. The pH of SOL-B was adjusted to 2. The suspension that formed was then aged (2 h) followed by hydrothermal treatment of the bottom yellow precipitate (200 °C, 24 h). The product that formed was washed (H_2O , $\text{C}_2\text{H}_5\text{OH}$), dried in a hot air oven (70 °C, 24 h), and assigned the name $\text{BiVO}_4\text{-H}$. SOL-C was stirred followed by slow addition of 10 g of $\text{CO}(\text{NH}_2)_2$. The suspension was refluxed (84 °C, 24 h). The precipitate that formed was then washed (H_2O , $\text{C}_2\text{H}_5\text{OH}$), dried in a hot air oven (70 °C, 24 h), and assigned the name $\text{BiVO}_4\text{-R}$. All of the fabricated samples were collected and ground well for further characterizations (Scheme 1).^{11,27,32,33}

Scheme 1. Illustration of Diverse Fabrication Techniques for Hierarchical Microstructures of Monoclinic Scheelite BiVO_4 Samples



2.3. Fabrication of Ag- $\text{BiVO}_4\text{-R}$. The successful deposition of Ag nanoparticles over the $\text{BiVO}_4\text{-R}$ sample was carried out via a photodeposition technique. In brief, 300 mg of the $\text{BiVO}_4\text{-R}$ sample was well dispersed in 35 mL of a $\text{DI-H}_2\text{O}$ /methanol mixture (4/1) via ultrasonication for 30 min. Subsequently, 3.3 mL of an AgNO_3 aqueous solution (10 g L^{-1}) was added dropwise to the above suspension with stirring for 30 min followed by visible-light irradiation for 2 h by a 300 W Xe lamp. The resulting suspension was then collected, washed ($\text{DI-H}_2\text{O}$, ethanol), dried in a hot air oven (60 °C for 12 h), and assigned the name Ag- $\text{BiVO}_4\text{-R}$. The fabricated yellowish green Ag- $\text{BiVO}_4\text{-R}$ sample was collected and ground well for further characterizations.^{27,29}

2.4. Analytical Characterizations. Crystallographic features were cautiously examined in a Rigaku Ultima IV X-ray diffractometer ($\text{Cu K}\alpha$, $\lambda = 1.5 \text{ \AA}$, $2\theta = 10\text{--}80^\circ$, 5° min^{-1} , 40 kV, 40 mA). A Zeiss Supra 55 field emission scanning electron microscope was employed to examine the microstructure and surface morphological information. Photophysical investigations were retrieved from a JASCO 750 UV–

visible diffuse reflectance spectrometer (BaSO_4 reference). Photoluminescence spectra were precisely examined by deploying a JASCO FP8300 spectrofluorometer ($\lambda_{\text{ex}} = 325 \text{ nm}$). A JASCO FT/IR-4600 Fourier transform infrared spectrometer was utilized to neatly analyze the bond vibrational chemistry (KBr matrix). Transmission electron microscopy and high-resolution transmission electron microscopy was carried out to study the surface morphology, deploying a Philips Tecnai G2 instrument (200 kV). The surface chemical characteristics of fabricated materials were investigated by a VG Microtech Multilab ESCA 3000 XPS instrument (Al $K\alpha$, 1486.7 eV, C 1s 284.9 eV).

2.5. Electrochemical Measurements. An IVIUM n STAT electrochemical workstation with a conventional three-electrode Pyrex electrochemical cell (working electrode Pt, Ag/AgCl, 0.1 M Na_2SO_4 , pH 6.1) was employed to examine photoelectrochemical data. Working electrodes were fabricated by an electrophoretic deposition method in a 20 mL acetone dispersion of 20 mg of the photocatalyst and 20 mg of iodine. The coating surface areas of FTOs (fluorine-doped tin oxide) were maintained to 1 cm \times 1 cm under a controlled potentiostatic bias (60 V, 5 min). Linear sweep voltammetry (LSV, -1 to $+1 \text{ V}$) analysis, electrochemical impedance spectroscopy (EIS, 0.1–10000 Hz, 0.63 V amplitude, 0 bias), and Bode measurements were executed under visible-light irradiation (300 W xenon, cut off filter $\geq 400 \text{ nm}$). The Mott–Schottky analysis was carried out in the dark (500 Hz, 25 mV). No photoresponse was observed for FTO without a catalyst coating. The photocurrent steadiness was measured by chronoamperometric (CA) analysis (600 s, 0.6 V). Transient photocurrent measurements were carried out under chopped visible-light illumination (300 s, 0.6 V).

2.6. Photocatalytic Levofloxacin Detoxification. A quartz reactor was employed to examine photocatalytic levofloxacin detoxification under irradiation of visible light (300 W Xe lamp, $\lambda \geq 420 \text{ nm}$, 100 mW/cm^2). Before photoirradiation, a suspension of 20 mg of the catalyst in 20 mL of a levofloxacin aqueous solution with 20 mg L^{-1} concentration was stirred magnetically in the dark for 30 min to enable an adsorption–desorption equilibrium. The resulting suspension was centrifuged and analyzed in a UV–vis spectrometer at 288 nm.^{34–36} The process was repeated at varying reaction times to evaluate the rate and the reaction kinetics.

2.7. Photocatalytic Water Oxidation Reaction. A sealed quartz batch reactor (100 mL capacity) with a temperature controller was employed to examine the water oxidation reaction under visible light irradiation (300 W Xe lamp, $\lambda \geq 420 \text{ nm}$). First a suspension of 20 mg of the catalyst in 20 mL of a 0.05 M AgNO_3 aqueous solution was purged with nitrogen for 30 min preirradiation at room temperature. The source was maintained 10 cm away from the suspension. A molecular sieve (5 Å) column GC-17A gas chromatography instrument with a thermal conductivity detector (TCD) was employed to characterize evolved O_2 .

3. RESULTS AND DISCUSSION

FESEM and HRTEM analyses were carried out to better investigate the microstructure and surface morphology of the fabricated samples (Figure 1). As shown in Figure 1a, the pristine $\text{BiVO}_4\text{-C}$ appears to have a dense microspherical structure composed of irregular-polyhedral crystals.^{11,23} The minuscule pores near the junction of nanopolyhedral structures in the microstructures are ascribed to gas amputation during the calcination process. The irregular Anonna reticulate shaped microstructures have an average particle size of 3.4–9.6 μm . The resultant TEM image of the fabricated $\text{BiVO}_4\text{-C}$ sample exhibits a microspherical morphology with a well-defined outline, which is in good agreement with the FESEM results (Figure 1b). Furthermore, a high-resolution FESEM image of $\text{BiVO}_4\text{-H}$ (Figure 1c) shows a truncated-bipyramidal morphology with a smooth surface and two {040}, four {110}, and four {011} well-exposed facets.^{27–29,33} The average particle size of $\text{BiVO}_4\text{-H}$ was evaluated to be 1.5–4 μm . The respective TEM

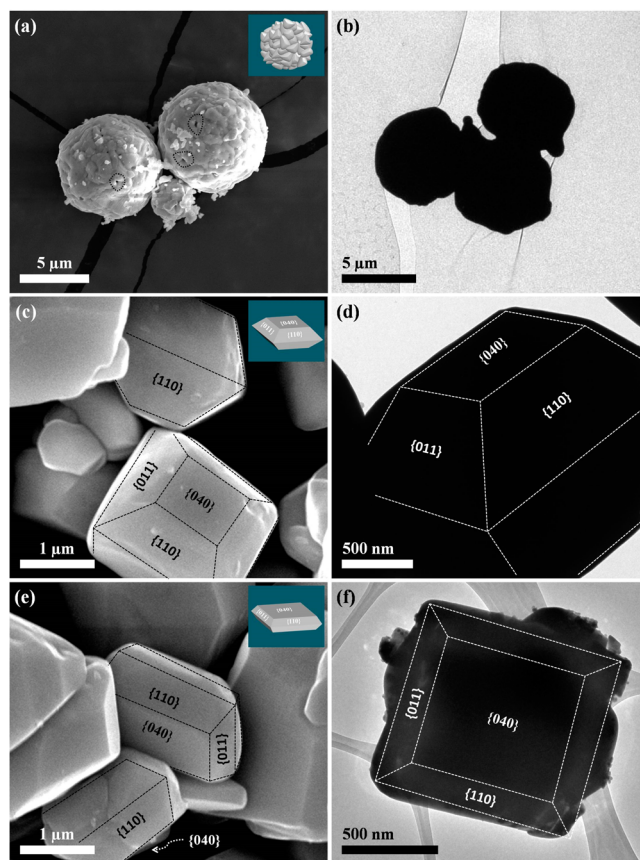


Figure 1. Respective FESEM and TEM images of fabricated monoclinic scheelite (a, b) $\text{BiVO}_4\text{-C}$, (c, d) $\text{BiVO}_4\text{-H}$, and (e, f) $\text{BiVO}_4\text{-R}$ samples.

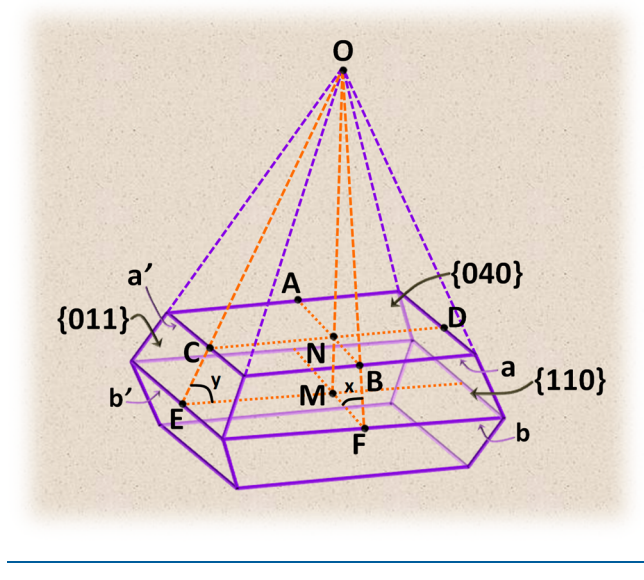
image of the $\text{BiVO}_4\text{-H}$ sample (Figure 1d) illustrates a truncated-bipyramidal morphology with well-defined outlines analogous to the respective FESEM images. Figure 1e represents the FESEM image of the $\text{BiVO}_4\text{-R}$ sample. Microparticles with a truncated-bipyramidal shape with two {040}, four {110}, and four {011} highly exposed facets are well observed.^{27–29,33} The approximate average particle size was calculated to be 1.5–5 μm . Expectedly, an analogous truncated-bipyramidal morphology with well-defined outlines was also observed for the $\text{BiVO}_4\text{-R}$ sample (Figure 1f).

The exposure of various {040}, {110}, and {011} facets affects the exciton separation ability and thus affects the overall redox efficiency of the photocatalysts. The {040} facet is composed of a BiV_4 multiatomic center, while the {110} and {011} facets are composed of Bi_4V_2 , thus suggesting analogous characteristics of {110} and {011} exposed facets.³³ The drift velocity of photoexcited electrons over the {040} facet was reported to be greater than that of the {110} facet due to the lower effective mass of electrons.^{27,28,30,37,38} Moreover, the densities of states near the valence band maximum (VB_{max}) and conduction band minimum (CB_{min}) of {011} is more localized in comparison to those for the {040} facet.^{27,28,30,38} Furthermore, lower band edge potentials of the {040} facet in comparison to the {110} facet favors thermodynamic exciton separations over the exposed facets of inbuilt {040/110} facet isotype heterojunctions.^{29,33,38} Consequently, photoexcited electrons tend to move over {040} reduction facets while causing the accumulation of photogenerated holes over {110} oxidation facets, thus promoting the development of built-in

{040/110} facet isotype heterojunctions over BiVO₄-H and BiVO₄-R samples.^{29,30,32,33,37}

In addition, the relative {040} facet exposure percentage of fabricated materials were well-evaluated via a slab-equilibrium model analysis (Scheme 2) to better validate the effective

Scheme 2. Schematic Illustration of the Slab-Equilibrium Model of a Monoclinic Scheelite BiVO₄ Single Crystal



involvement of the exposure ratio toward enhanced activity.^{39–41} As illustrated in Figure 1, the BiVO₄-C microstructures show no obvious high exposure of {040}, {110}, and {011} active functional facets. However, BiVO₄-H and BiVO₄-R exhibit well-exposed {040} facets with an average exposure values of 14% and 36%, respectively, thus validating a higher exposure factor of the {040} facet ({040}/{110}) in BiVO₄-R in comparison to BiVO₄-C and BiVO₄-H. This is further explored in the discussion below.

The theoretical assessment of {040} facet exposure percentage was carried out by employing the equations^{39–41}

$$A_{\{040\}} = 2aa' \quad (1)$$

$$A_{\{110\}} = 4\left(\frac{1}{2}OF \times b - \frac{1}{2}OB \times a\right) \quad (2)$$

$$A_{\{011\}} = 4\left(\frac{1}{2}OE \times b' - \frac{1}{2}OC \times a'\right) \quad (3)$$

$$A_{\{040\}} (\%) = \frac{A_{\{040\}}}{A_{\{040\}} + A_{\{110\}} + A_{\{011\}}} \times 100 \quad (4)$$

$$A_{\{040\}} (\%) = \frac{2aa'}{\left\{2aa' + 4\left(\frac{1}{2}OF \times b - \frac{1}{2}OB \times a\right) + 4\left(\frac{1}{2}OE \times b' - \frac{1}{2}OC \times a'\right)\right\}} \times 100 \quad (5)$$

$$A_{\{040\}} (\%) =$$

$$\frac{2aa'}{\left\{2aa' + 4\left(\frac{1}{2}\frac{b'}{\cos x}b - \frac{1}{2}\frac{a'}{\cos x}a\right) + 4\left(\frac{1}{2}\frac{b'}{\cos y}b' - \frac{1}{2}\frac{a'}{\cos y}a'\right)\right\}} \times 100 \quad (6)$$

$$A_{\{040\}} (\%) = \frac{aa'}{\left\{aa' + \frac{1}{2}(bb' - aa')\left(\frac{1}{\cos x} + \frac{1}{\cos y}\right)\right\}} \times 100 \quad (7)$$

Here, $A_{\{040\}}$, $A_{\{110\}}$, and $A_{\{011\}}$ denote the exposure areas of {040}, {110}, and {011} functional facets of the BiVO₄ {040/110} facet isotype heterojunction, respectively. a and a' signify the length and width of the well-exposed {040} facet,

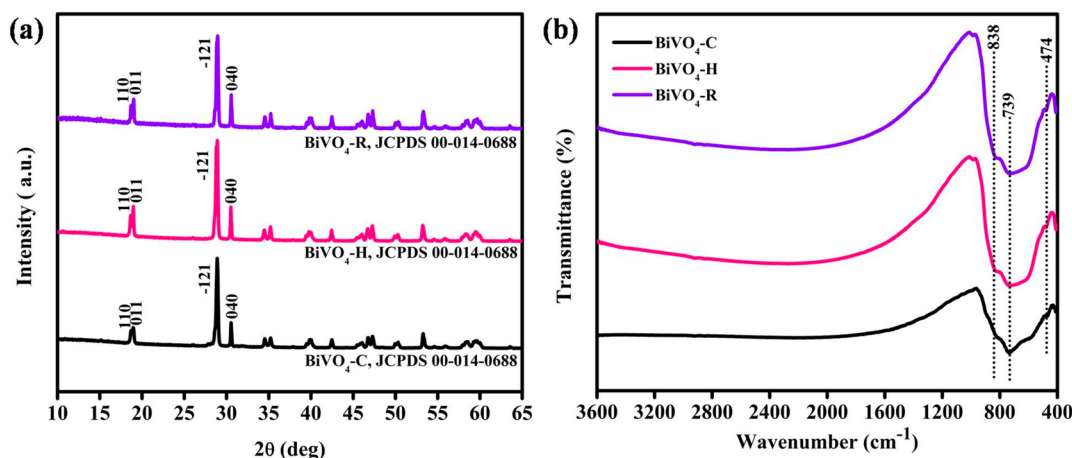


Figure 2. Comparative (a) X-ray diffraction patterns and (b) FT-IR spectra of fabricated monoclinic scheelite BiVO₄ samples.

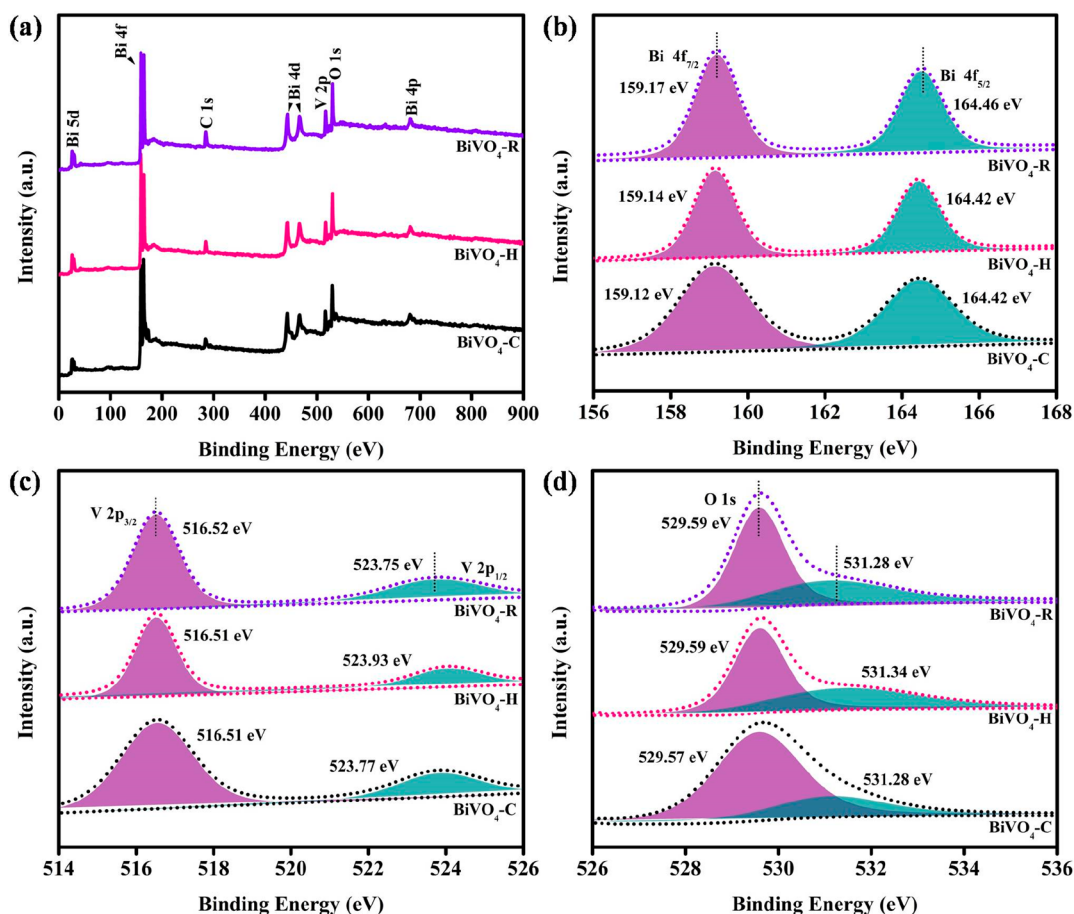


Figure 3. Comparative XPS analysis overview: (a) survey spectrum and (b) Bi 4f, (c) V 2p, and (d) O 1s narrow scan spectra of fabricated monoclinic scheelite BiVO_4 samples.

respectively. b and b' signify the length and width of the truncated bipyramid, respectively. x ($\sim 66.06^\circ$) and y ($\sim 66.48^\circ$) are the theoretical inclinations of $\{110\}$ and $\{011\}$ facets to the $\{040\}$ facet, respectively.

Powder XRD analysis providing particulars of crystallographic features of the as-synthesized catalysts are shown in Figure 2a. The elemental diffraction peaks of the pristine BiVO_4 -C, BiVO_4 -H, and BiVO_4 -R are well indexed with a body-centered monoclinic scheelite phase of BiVO_4 with JCPDS Card No. 00-014-0688.^{11,27–29,32,33} The obtained diffraction spectra were devoid of any ancillary impurity peaks, suggesting high purity of the catalysts. Furthermore, the decidedly crystalline characteristics of the pristine synthesized samples were well established by the sharp and intense diffraction signals.

Moreover, the crystallite sizes of the as-synthesized pristine materials were estimated to be 38.88, 64.59, and 64.09 nm for BiVO_4 -C $\{112\}$, BiVO_4 -H $\{-121\}$ and BiVO_4 -R $\{-121\}$, respectively, by employing the Scherrer equation (eq 8):^{11,26}

$$\text{crystallite size } (D) = k\lambda/\beta \cos \theta \quad (8)$$

Here, λ , β , and θ denote the irradiation wavelength, half-maximum peak width, and diffraction angle of the diffraction peak under inspection, respectively. Moreover, k denotes the Scherrer constant with a value of 0.94. It is worth noting that the intensities of characteristic diffraction signals of the fabricated samples follow the order BiVO_4 -C $\{-121\}$ < BiVO_4 -R $\{-121\}$ < BiVO_4 -H $\{-121\}$. However, the $\{040\}$ /

$\{110\}$ diffraction signal intensity ratio is shown to be elevated in BiVO_4 -R relative to BiVO_4 -C and BiVO_4 -H, suggesting superior growth of BiVO_4 crystals vertical to the $\{110\}$ facets.³³ This was further attested by FESEM analysis. Moreover, a minor variation in fwhm value was observed, which suggests the existence of a minor distortion of the $[\text{VO}_4]^{3-}$ tetrahedron in the local structure.²⁶

Furthermore, broad and readily distinguishable characteristic ν_3 stretching vibrational absorption spectra were observed in the range of 1000 to 500 cm^{-1} in FT-IR analyses of fabricated BiVO_4 samples (Figure 2b). The characteristic signals at around 739 and 838 cm^{-1} are well-attributed to ν_3 asymmetric and ν_1 symmetric stretching vibrational modes of the VO_4^{3-} unit, respectively.⁴² Moreover, a weak signal around 474 cm^{-1} was observed and ascribed to the ν_4 bending vibrational mode of VO_4^{3-} .⁴³ This is in good agreement with the monoclinic scheelite phase of BiVO_4 samples.

Furthermore, the valence band of monoclinic scheelite BiVO_4 is reported to be composed of Bi_{6s} and O_{2p} hybrid orbitals, while the conduction band is composed of a V_{3d} orbital.^{11,23} The existence of distortion of the VO_4^{3-} tetrahedron in the local structure along with superior crystallinity is responsible for escalating the overlap of Bi_{6s} and O_{2p} orbitals of monoclinic scheelite BiVO_4 . This consequently enhanced the photoexciton delocalization process and enhanced the migration of photogenerated holes to the surface, accordingly resulting in a superior water oxidation reaction.^{24,26} Thus, when the X-ray diffraction facts

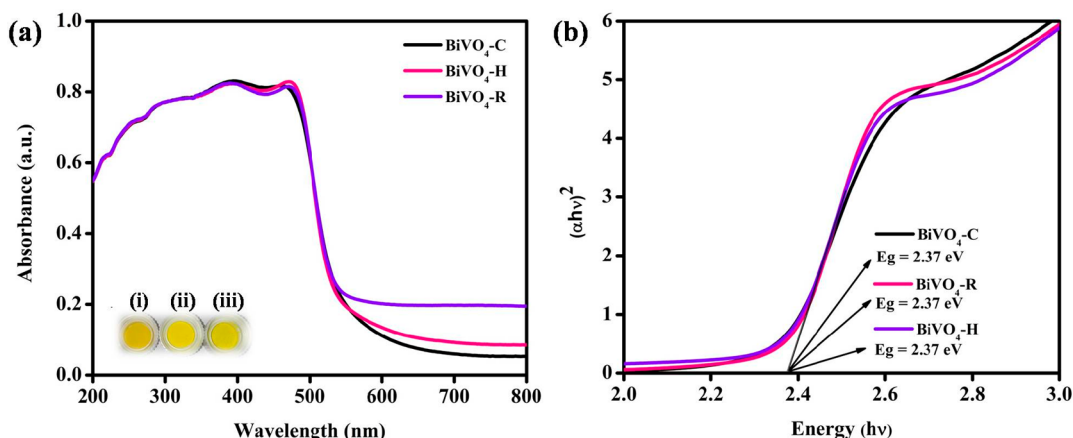


Figure 4. (a) UV-vis DRS spectra and (b) Tauc plots of fabricated monoclinic scheelite BiVO_4 samples.

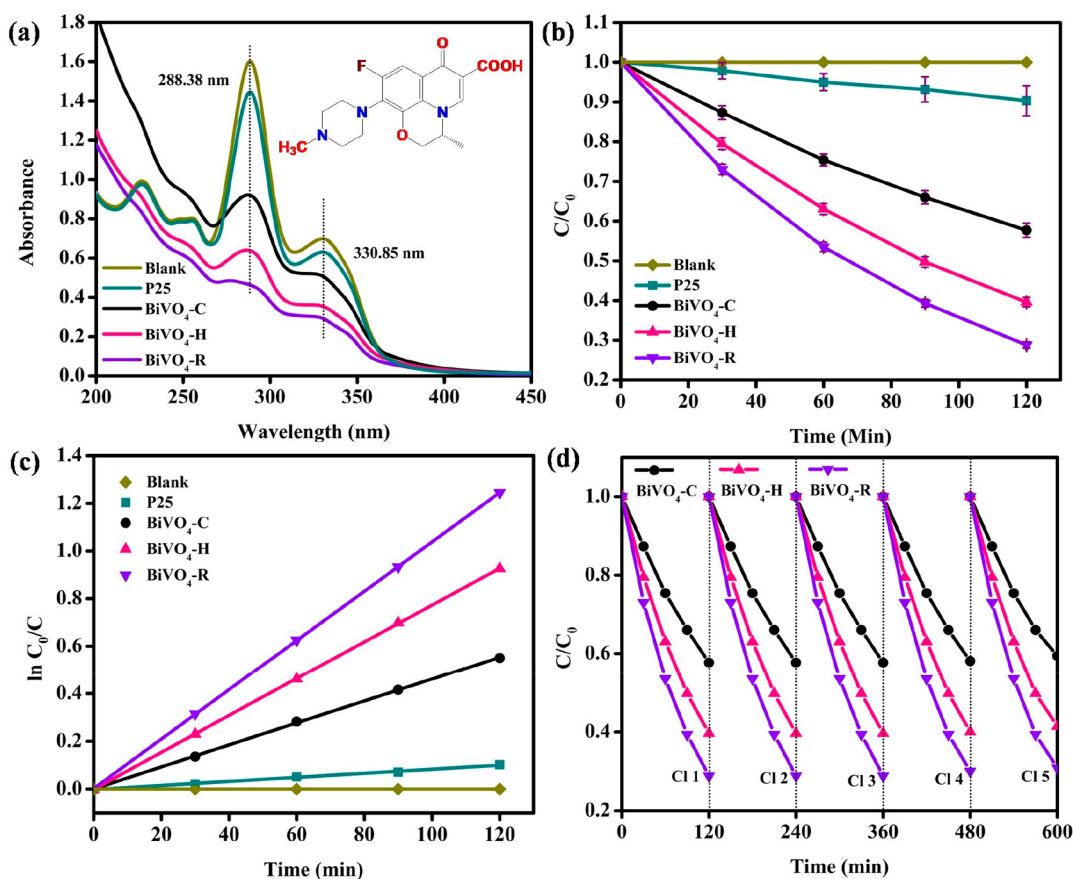


Figure 5. (a) UV-vis absorption spectrum of levofloxacin undergoing photodetoxification, (b) detoxification over time, (c) kinetics of detoxification, and (d) cyclic reusability measurements over fabricated monoclinic scheelite BiVO_4 samples.

are taken into account, the higher crystallinity of $\text{BiVO}_4\text{-H}$ attests to its superior photocatalytic redox activity over $\text{BiVO}_4\text{-R}$ and $\text{BiVO}_4\text{-C}$ sequentially. This was further supported by the crystallite sizes of fabricated catalysts. The larger crystallite size of $\text{BiVO}_4\text{-H}$ suggests a higher photocatalytic activity in comparison to $\text{BiVO}_4\text{-R}$ and $\text{BiVO}_4\text{-C}$.^{24,26}

XPS measurements of fabricated hierarchical structures of the BiVO_4 material were carried out to thoroughly investigate the surface elemental composition and valence states (Figure 3). The XPS survey spectrum of fabricated BiVO_4 materials illustrates the presence of Bi, V, and O devoid of any adulteration (Figure 3a). The appearance of an intense XPS

signal of C near 284.9 eV was readily attributed to the adventitious hydrocarbon.^{11,12} Two prominent XPS signals are observed for Bi 4f narrow scan spectrum for the fabricated $\text{BiVO}_4\text{-C}$ (159.12, 164.42 eV), $\text{BiVO}_4\text{-H}$ (159.14, 164.42 eV), and $\text{BiVO}_4\text{-R}$ (159.17, 164.46 eV) materials. The observed binding energy differences of 5.3, 5.28, and 5.29 eV are due to the respective Bi 4f_{7/2} and Bi 4f_{5/2} spin-orbit splitting and +3 state of the Bi atom (Figure 3b).^{11,12} The characteristic spin-orbit splitting of V 2p_{3/2} and V 2p_{1/2} were observed for fabricated $\text{BiVO}_4\text{-C}$ (516.51, 523.77 eV), $\text{BiVO}_4\text{-H}$ (516.51, 523.93 eV) and $\text{BiVO}_4\text{-R}$ (516.52, 523.75 eV) materials with binding energy differences of 7.26, 7.42, and 7.23 eV,

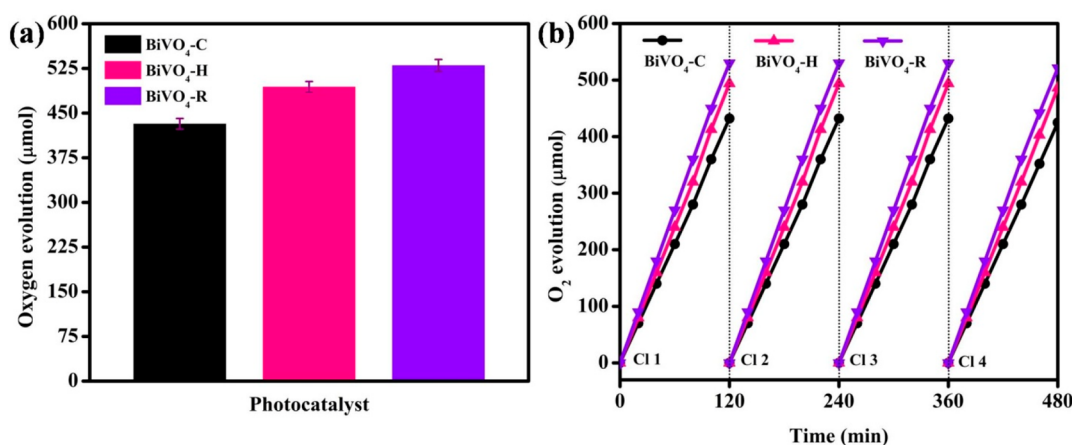


Figure 6. Comparative (a) O_2 evolution and (b) cyclic reusability measurements over fabricated monoclinic scheelite BiVO_4 samples.

respectively, ascribed to the +5 valence state of the vanadium atom (Figure 3c).^{11,12} Moreover, the O 1s XPS spectra of fabricated $\text{BiVO}_4\text{-C}$ (529.57, 531.28 eV), $\text{BiVO}_4\text{-H}$ (529.59, 531.34 eV), and $\text{BiVO}_4\text{-R}$ (529.59, 531.28 eV) materials illustrate a pair of intense signals corresponding to lattice oxygens of Bi–O and V–O bonds (lower BE energy) and hydroxyl oxygen (–OH) (higher BE energy) (Figure 3d).²² The results illustrate the existence of a BiVO_4 material with a monoclinic scheelite crystalline phase.

The strong optoelectronic response of the pristine BiVO_4 catalysts with strapping (strong) absorption characteristic in visible region was well exposed by UV–vis diffuse reflectance spectroscopy (Figure 4a). The observed absorption edges were devoid of significant differences with approximate values of nearly 551–554 nm for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively, with a trifling red shift of $\text{BiVO}_4\text{-C}$ in comparison to the other pristine catalysts.³³ The observed absorption characteristics were consistent with the closely analogous colors of fabricated (i) $\text{BiVO}_4\text{-C}$, (ii) $\text{BiVO}_4\text{-H}$, and (iii) $\text{BiVO}_4\text{-R}$ samples (Figure 4a, inset). Moreover, the absorption edges were projected to originate from respective optical band-gap transitions rather than from any impurity levels. The close band-gap values were estimated by implementing eq 9:^{11,16,44}

$$\alpha h\nu = A(h\nu - E_g)^{n/2} \quad (9)$$

Here, α , h , A , ν , and E_g denote the absorption coefficient, Planck's constant, the proportionality constant, light frequency, and band-gap energy, respectively. The value of n was adopted as 1 for the direct transition natures of fabricated samples.^{11,16} The probable close band-gap energies were observed to have no apparent divergences and evaluated to be 2.37 eV for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively (Figure 4b), which are in accordance with the absorption analysis and previous reports.^{11,28} Moreover, the light-harvesting efficiency was proposed to have no effective association toward the diverse photocatalytic efficiencies of fabricated shape-selective BiVO_4 materials.

Furthermore, a comparison of the photoredox efficiencies of the fabricated catalysts were investigated by performing degradation of an organic deleterious waste levofloxacin ($\text{C}_{18}\text{H}_{20}\text{FN}_3\text{O}_4$) aqueous solution over 120 min of visible light illumination. For a comparative illustration commercial-grade Degussa P25 was also introduced. A drastic decrease in the respective signal intensities for aromatic (288.38 nm) and

piperazine (330.85 nm) rings in the UV–vis absorption spectrum evidenced the photocatalytic degradation process (Figure 5a, inset).^{23,35} Negligible detoxification results were observed for Degussa P25, while roughly no degradation was observed for a solution lacking a catalyst (photolysis). The degradation percentages of levofloxacin were evaluated to be 42.3%, 60.4%, and 71.2% for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively. Moreover, the detoxification was well observed by employing a pseudo-first-order kinetic model (Figure 5b,c) and the detoxification rate constant was evaluated adopting eq 10:^{11,17}

$$\ln(C/C_0) = -kt \quad (10)$$

Here, C , C_0 , k , and t denote the final concentration, the initial concentration, the rate constant, and the reaction time, respectively. A higher rate constant for $\text{BiVO}_4\text{-R}$ (0.0104 min^{-1}) was apparent, which is 1.35- and 2.24-fold superior to those of the fabricated $\text{BiVO}_4\text{-H}$ (0.0077 min^{-1}) and $\text{BiVO}_4\text{-C}$ (0.0046 min^{-1}) samples.

Cyclic reuses of fabricated BiVO_4 hierarchical microstructures were carried out to investigate the apparent photostability and recyclability (Figure 5d). The detoxification was observed to be consistent up to four consecutive cycles with minimal deteriorations of 1.82%, 1.86%, and 2.03% in the fifth cycle for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively, ascribed to the wear and tear of the catalyst surface and inevitable losses during the intermediary recovery process.^{11,12,22}

In addition, photocatalytic oxygen evolution efficiencies of fabricated samples were carried out in AgNO_3 aqueous solution under visible-light illumination for 120 min. Expectedly, $\text{BiVO}_4\text{-R}$ illustrated an elevated water oxidation efficiency of 530.6 μmol , which was about 1.22 and 1.07 times those of $\text{BiVO}_4\text{-C}$ (434.2 μmol) and $\text{BiVO}_4\text{-H}$ (494.8 μmol), respectively (Figure 6a). Cyclic reuses of the fabricated hierarchical $\text{BiVO}_4\text{-R}$ sample showed significant steadiness up to three successive cycles followed by deteriorations of 1.64%, 1.58%, and 1.62% in the fourth cycle for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively. The inconsistency observed was ascribed to the wear and tear of the catalyst surface over the reaction period. Moreover, the inescapable metallic Ag cluster deposition owing to photoreduction of Ag^+ ions over the catalyst surface consequently hinders visible light penetration, thus retarding the activity (Figure 6b).^{11,12,22} The diverse photocatalytic response was credited to varied photo electron–

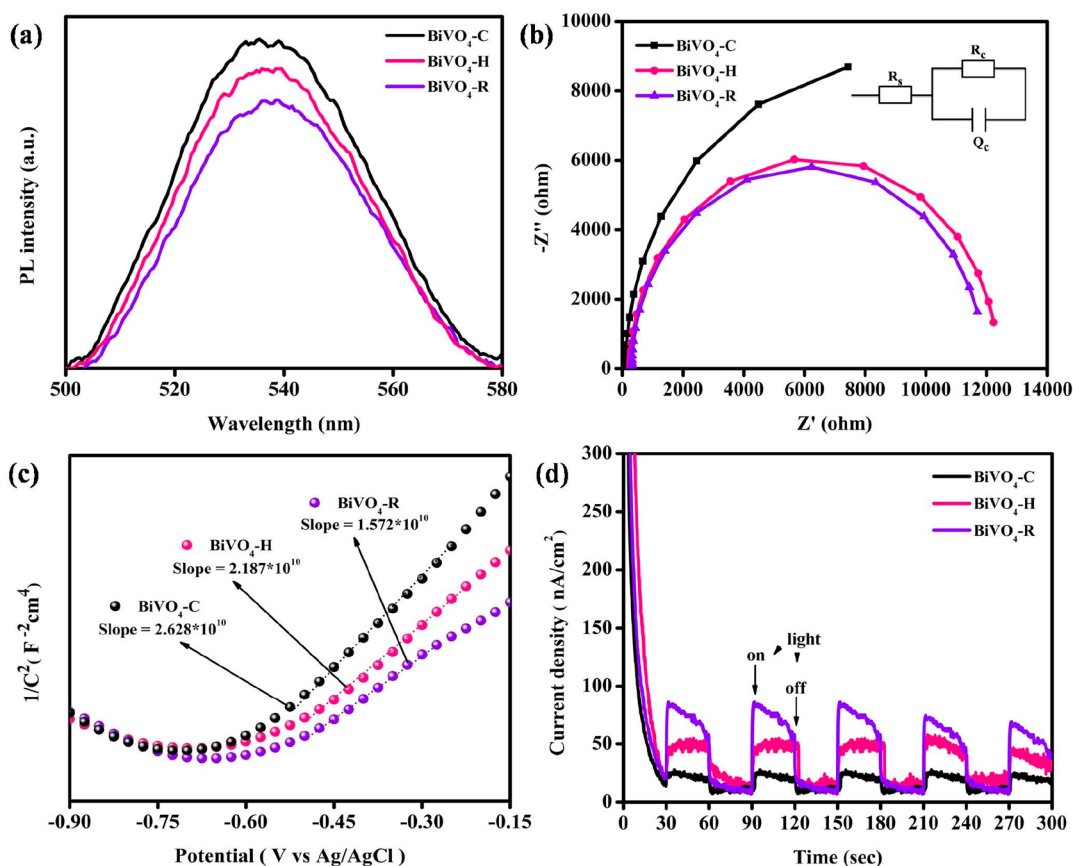


Figure 7. (a) PL, (b) EIS, (c) carrier density, and (d) transient photocurrent responses of fabricated monoclinic scheelite BiVO_4 samples.

hole anti-recombination processes occurring in the crystal microstructures and was well-explored via photoelectrochemical measurements.

Moreover, a detailed investigation regarding photoexciton separation, migration, and recombination in the fabricated materials was well established via PL analysis ($\lambda_{\text{ex}} = 325$ nm, room temperature). As shown in Figure 7a, the broad PL spectra of the fabricated catalysts are centered at 536 nm and are ascribed to the corresponding band-gap emissions.^{11,27} Expectedly, PL intensities of the catalysts had minimal differences. Nevertheless, $\text{BiVO}_4\text{-R}$ has a comparatively lower PL intensity followed by $\text{BiVO}_4\text{-H}$ and $\text{BiVO}_4\text{-C}$, suggesting lower charge carrier recombination rates.^{11,27} This was further corroborated by the electrochemical impedance analysis shown in Figure 7b. The exposed characteristic semicircular Nyquist plots validate the occurrence of exciton separation and transfer in fabricated samples.^{11,45} Moreover, a comparatively low frequency and smaller arc radius confirms an effective decrease in interfacial charge transfer resistance (R_c) in the fabricated $\text{BiVO}_4\text{-R}$ microstructure in comparison to the pristine $\text{BiVO}_4\text{-C}$ and $\text{BiVO}_4\text{-H}$. However, the variation was minimal in comparison to $\text{BiVO}_4\text{-H}$. The fitted Randle circuit model further validates a notable decrease in charge transfer resistance over the electrode–electrolyte interface of $\text{BiVO}_4\text{-R}$ ($11620 \Omega \text{ cm}^{-2}$) relative to $\text{BiVO}_4\text{-C}$ ($17640 \Omega \text{ cm}^{-2}$) and $\text{BiVO}_4\text{-H}$ ($12120 \Omega \text{ cm}^{-2}$) and thus illustrates possible superior charge carrier migration in architected $\text{BiVO}_4\text{-R}$ relative to $\text{BiVO}_4\text{-C}$ and $\text{BiVO}_4\text{-H}$ samples. R_s and Q_c elements (Figure 7b, inset) in the equivalent circuit denote series resistance and capacitance phase elements, respectively.^{11,45,46} Moreover, a lower positive slope was observed for $\text{BiVO}_4\text{-R}$

(1.572×10^{10}) relative to $\text{BiVO}_4\text{-H}$ (2.187×10^{10}) and $\text{BiVO}_4\text{-C}$ (2.628×10^{10}) sequentially (Figure 7c). This further indicates a higher carrier density (N_d) and thus elevated electrical conductance provoking the superior charge-transfer characteristics of $\text{BiVO}_4\text{-R}$ in comparison to the fabricated $\text{BiVO}_4\text{-C}$ and $\text{BiVO}_4\text{-H}$ samples. Moreover, the probable values of carrier densities (N_d) of fabricated BiVO_4 samples could be calculated by adopting eq 11:^{11,46–50}

$$\text{carrier density } (N_d) = (2/e\epsilon\epsilon_0)B \quad (11)$$

Here, e , ϵ , ϵ_0 , and B denote the electron charge (1.602×10^{-19} C), the relative dielectric constant of the semiconducting material, the vacuum permittivity ($8.85 \times 10^{-12} \text{ F m}^{-1}$), and the slope value of MS plot tangents, respectively. Consequently, an increased carrier concentration will lead to elevated charge-transfer characteristics of BiVO_4 as per eq 12:⁵⁰

$$\text{electrical conductivity } (\sigma) = en\mu \quad (12)$$

Here, e , n , and μ denote the electronic charge unit (1.602×10^{-19} C), the charge carrier concentration, and the charge carrier mobility, respectively.

Moreover, the photoelectrochemical transient measurements illustrated an enhanced transient photocurrent intensity of $\text{BiVO}_4\text{-R}$ (Figure 7d). The results were observed to follow the trend of $\text{BiVO}_4\text{-C} < \text{BiVO}_4\text{-H} < \text{BiVO}_4\text{-R}$ for each consecutive on–off cycle, suggesting an improved exciton pair separation tendency and surface charge antirecombination process over $\text{BiVO}_4\text{-R}$ in comparison to the other pristine materials.^{11,22,51}

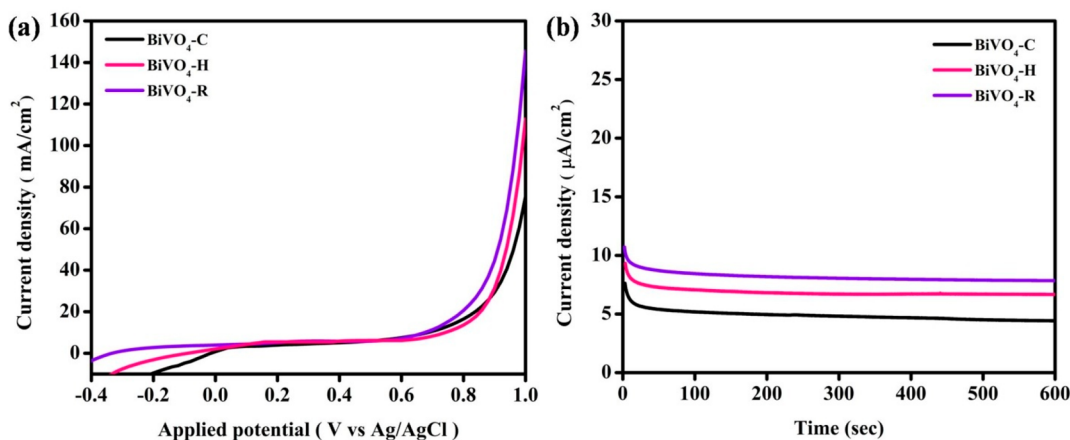


Figure 8. Comparative (a) LSV and (b) CA analysis responses of fabricated monoclinic scheelite BiVO_4 samples.

Moreover, linear sweep voltammograms (LSV) exposing a higher photocurrent density of $\text{BiVO}_4\text{-R}$ ($145.6 \mu\text{A}/\text{cm}^2$) relative to $\text{BiVO}_4\text{-C}$ ($75.5 \mu\text{A}/\text{cm}^2$) and $\text{BiVO}_4\text{-H}$ ($113.1 \mu\text{A}/\text{cm}^2$) validates the sturdy charge carrier separation occurring over $\text{BiVO}_4\text{-R}$. Nevertheless, a drastic reduction in charge carrier recombination was validated by a notable negative shift in onset potential for $\text{BiVO}_4\text{-R}$ (-0.33 V) in comparison to the fabricated $\text{BiVO}_4\text{-C}$ (-0.013 V) and $\text{BiVO}_4\text{-H}$ (-0.11 V).⁵¹ The result is in good agreement with PL, impedance, carrier density, and transient photocurrent analysis (Figure 8a).

The time-variant photocurrent steadiness of the fabricated BiVO_4 materials was investigated by chronoamperometric analysis (Figure 8b). Expectedly, $\text{BiVO}_4\text{-R}$ illustrated an enhanced CA current of $7.9 \mu\text{A}$ with consistency up to 600 s. Following a minimal decay of 25.2% initially, there was no noticeable photocurrent loss observed over time, suggesting an enhanced electron lifetime in the $\text{BiVO}_4\text{-R}$ sample in comparison to $\text{BiVO}_4\text{-C}$ and $\text{BiVO}_4\text{-H}$. Moreover, the measured CA results followed the trend $\text{BiVO}_4\text{-C}$ (42% , $4.52 \mu\text{A}$) < $\text{BiVO}_4\text{-H}$ (28% , $6.7 \mu\text{A}$) < $\text{BiVO}_4\text{-R}$. The diverse photoelectrochemical responses were quite consistent with the observed photocatalytic outcomes.

Furthermore, a Bode analysis was executed to better understand the hole relaxation efficiencies of fabricated samples. A relatively positive shift of the frequency phase peak of $\text{BiVO}_4\text{-H}$ in comparison to $\text{BiVO}_4\text{-R}$ and $\text{BiVO}_4\text{-C}$ as projected in Bode plots reveals a decrease in hole separation/relaxation lifetime (τ_p) for $\text{BiVO}_4\text{-H}$ in comparison to the other pristine phases (Figure 9). Moreover, the entire separation/relaxation lifetimes in the fabricated BiVO_4 samples were evaluated to be 0.91, 0.48, and 0.61 ms for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively, adopting eq 13:^{51–54}

$$\text{hole relaxation lifetime } (\tau_p) = 1/2\pi f \quad (13)$$

Here, f denotes the relaxation frequency obtained from the maximum value allied with the Bode frequency range. A decrease in hole relaxation lifetime suggests a decrease in hole diffusion length (L_D) and thus a rapid hole diffusion process. The diffusion lengths were evaluated to be 2.17, 1.57, and 1.78 μm for $\text{BiVO}_4\text{-C}$, $\text{BiVO}_4\text{-H}$, and $\text{BiVO}_4\text{-R}$, respectively, adopting eq 14:^{51–54}

$$\text{hole diffusion length } (L_D) = (D\tau_p)^{1/2} \quad (14)$$

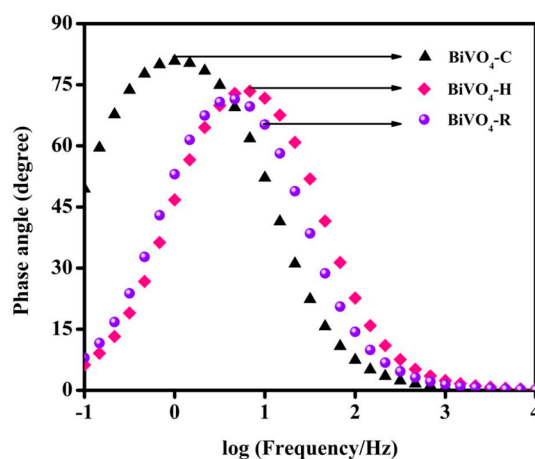
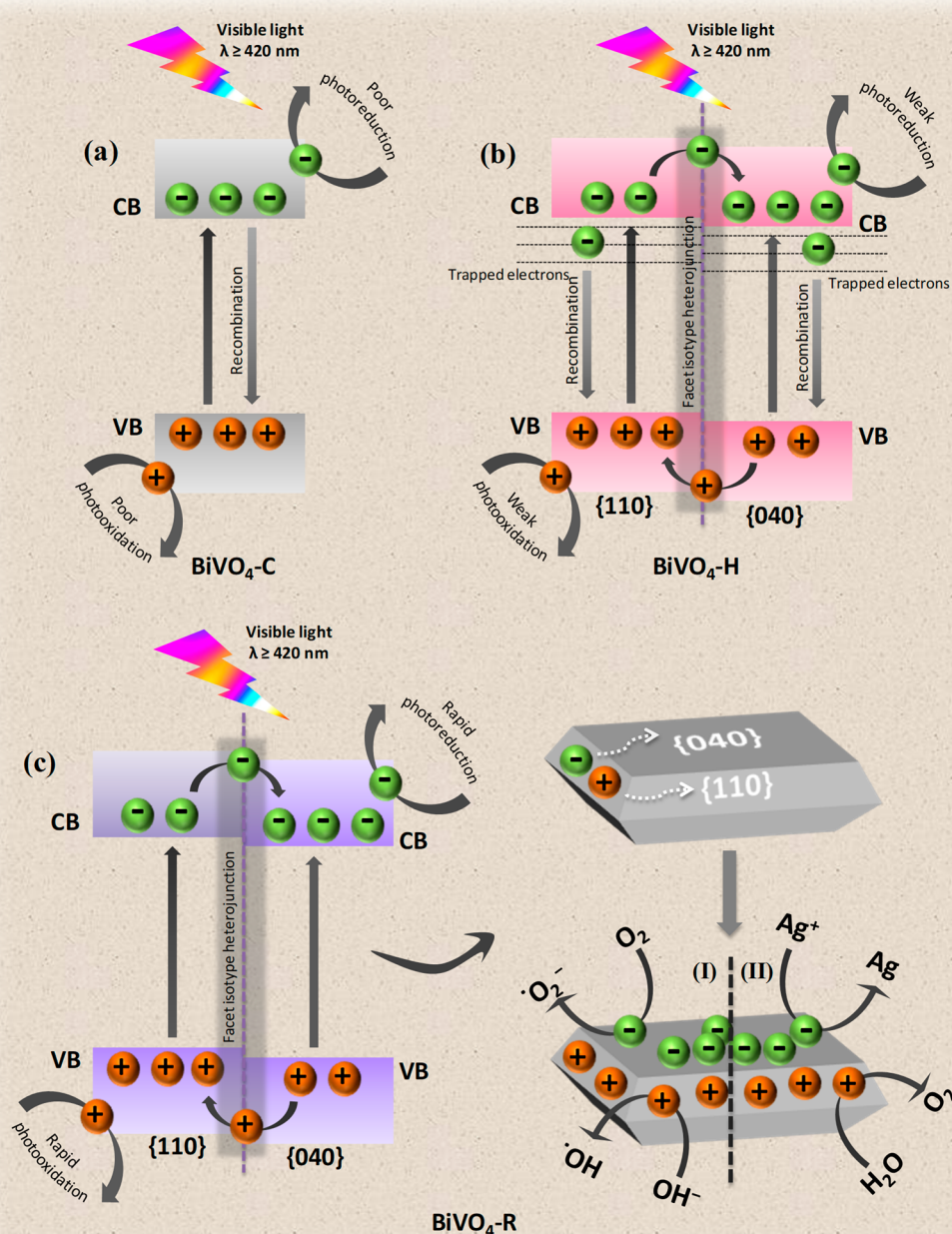


Figure 9. Bode analysis results of fabricated monoclinic scheelite BiVO_4 samples.

Here, D connotes the hole diffusion coefficient with a value of $5.2 \times 10^{-2} \text{ cm}^2 \text{ s}^{-1}$.²⁴ A reasonably lower hole diffusion length of $\text{BiVO}_4\text{-H}$ (1.57 μm) in comparison to $\text{BiVO}_4\text{-C}$ (2.17 μm) and $\text{BiVO}_4\text{-R}$ (1.78 μm) was observed, suggesting a superior hole consumed water decomposition reaction of $\text{BiVO}_4\text{-H}$ relative to $\text{BiVO}_4\text{-R}$ and $\text{BiVO}_4\text{-C}$ sequentially.^{24–26} This was further supported by crystallographic parameters and comparatively higher amounts of exposed {110} oxidation facets of the $\text{BiVO}_4\text{-H}$ sample.^{31,37} However, the constructive effects of the {040/110} facet isotype heterojunction and exposure ratio of various reduction–oxidation facets of monoclinic scheelite BiVO_4 were investigated to regulate the overall efficiencies of the catalysts.

As revealed by FESEM and TEM analysis, the fabricated $\text{BiVO}_4\text{-C}$ is devoid of any good exposure extent, while $\text{BiVO}_4\text{-H}$ and $\text{BiVO}_4\text{-R}$ samples have well-exposed {040} reduction and {110} oxidation facets and built-in {040/110} facet isotype heterojunctions.^{27–29} With no perceptible exposed redox facets, $\text{BiVO}_4\text{-C}$ suffers rapid exciton recombination (Scheme 3a). As revealed from XRD, FESEM, and TEM analysis, the {040/110} exposure factor in $\text{BiVO}_4\text{-H}$ is noticeably lower relative to $\text{BiVO}_4\text{-R}$. Reasonably more exposed and energetically favorable water oxidation {110} facets suggest higher hole provoked efficiency of $\text{BiVO}_4\text{-H}$; however, recombination occurring due to trapped electrons owing to minor exposed {040} facets decrease the activity

Scheme 3. Graphics of Photoexciton Separation/Recombination Phenomena over (a) $\text{BiVO}_4\text{-C}$ and (b) $\text{BiVO}_4\text{-H}$ {040/110} Facet Isotype Heterojunctions and (c) Mechanistic Pathway of Exciton Separation over a $\text{BiVO}_4\text{-R}$ {040/110} Facet Isotype Heterojunction with Photocatalytic (I) Levofloxacin Detoxification and (II) Water Oxidation



(Scheme 3b).³¹ Moreover, a higher {110} facet exposure causes an electron overflow effect provoked retardation of electron transfer to the {040} facet and thus validates the existence of interior exciton pair recombination.³⁹

In contrast, highly exposed {040} facets have been reported to possess a superior water molecule adsorption efficiency due to the promotion of the local bonding configuration along with the presence of multiatomic BiV_4 centers responsible for electron transfer for water oxidation (Scheme 3c).^{29,55} Thus, an increase in the relative exposure factor of {040} facet ({040}/{110}), i.e. a greater exposed {040} exposure percentage, possesses a constructive effect toward exciton separation and anti-recombination processes over {040/110}

facet isotype heterojunctions and hence superior photocatalytic activity.^{33,38,56} Consequently, the phenomenon results in an enhanced effective separation of photoexcitons and redox efficiency over $\text{BiVO}_4\text{-R}$ in comparison to $\text{BiVO}_4\text{-H}$ and $\text{BiVO}_4\text{-C}$, which is in good agreement with the results of photocatalytic and photoelectrochemical measurements.

Furthermore, the inbuilt {040/110} facet isotype heterojunction provoked augmented accumulation of photoelectrons over the {040} reduction facet of $\text{BiVO}_4\text{-R}$ was further evidenced by introducing Ag nanoparticles via a photo-deposition method. A powder XRD spectrum of the Ag- $\text{BiVO}_4\text{-R}$ sample shows the existence of {111} and {200} crystal plane reflections (JCPDS 00-001-1167), which suggests

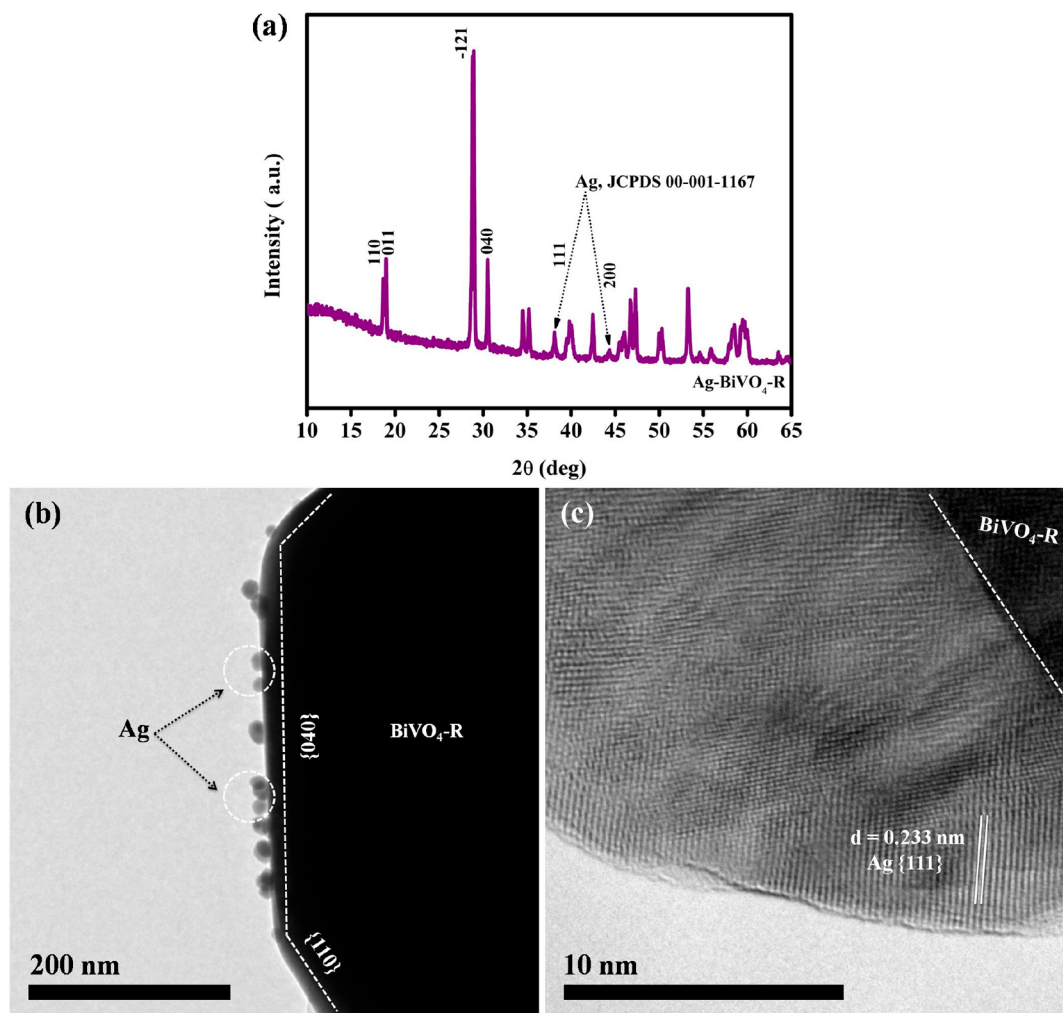
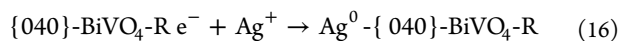
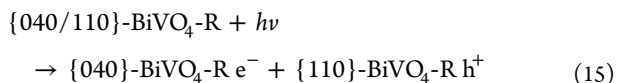


Figure 10. (a) XRD, (b) TEM, and (c) HRTEM results of the fabricated Ag-BiVO₄-R sample.

successful integration of metallic Ag with the BiVO₄-R sample (Figure 10a). A low-magnification TEM image of Ag-BiVO₄-R illustrates the augmented deposition of Ag nanoparticles over the {040} crystal facet of BiVO₄-R (Figure 10b). An observed interplanar spacing (*d*) of 0.233 nm was allied with the reflection of the {111} plane of metallic silver nanoparticles (Figure 10c). The deposition was ascribed to rapid-selective photoreduction of Ag⁺ particles to Ag⁰ via augmented photoelectrons accumulating over the {040} reduction facet of the BiVO₄-R {040/110} facet isotype heterojunction. The possible mechanistic pathway for the photodeposition is as follows:^{27,29,30}



A Mott–Schottky analysis was adopted to evaluate the probable conduction band edge potentials and charge-transfer characteristics of the fabricated monoclinic scheelite BiVO₄ samples (Figure 11a). The probable flat-band potentials (*E*_{fb}) were calculated to be −0.71, −0.68, and −0.63 V vs Ag/AgCl for BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively, from the *x* intercept of the extrapolated linear region of the *C*^{−2} versus potential plot employing eq 17.⁵⁶ Moreover, the relatively

positive shift in the flat band potential of BiVO₄-R suggests a minor electron-trapping process and thus an effectual lowering of photoelectron diffusion length from the bulk to the well-exposed {040} surface.³¹

$$1/C^2 = (2/e\epsilon\epsilon_0 N_d)[E - E_{fb} - (kT/e)] \quad (17)$$

Here, *C*, *e*, *ε*, *ε*₀, *N*_d, *E*, and *E*_{fb} denote the charge capacitance of free space, the charge of the elementary electron, the dielectric constant of the electrode material, the permittivity of free space, the electron donor density, the applied bias potential, and the flat band potential, respectively. Moreover, n-type characteristics of fabricated monoclinic BiVO₄ materials were well attested from positive slopes in the MS plot and probable conduction band (CB) values were calculated to be −0.15, −0.12, and −0.07 eV vs NHE for BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively. The result was due to the fact that *E*_{fb} of n-type materials gives an approximation of the conduction band minimum.^{11,57} Nevertheless, the valence band (VB) edges were evaluated to be 2.22, 2.25, and 2.30 eV vs NHE for BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively, adopting eq 18:⁵⁸

$$E_{CB} = E_{VB} - E_g \quad (18)$$

Here, *E*_{CB}, *E*_{VB}, and *E*_g denote the CB minimum, VB maximum, and band-gap energy, respectively. The relative

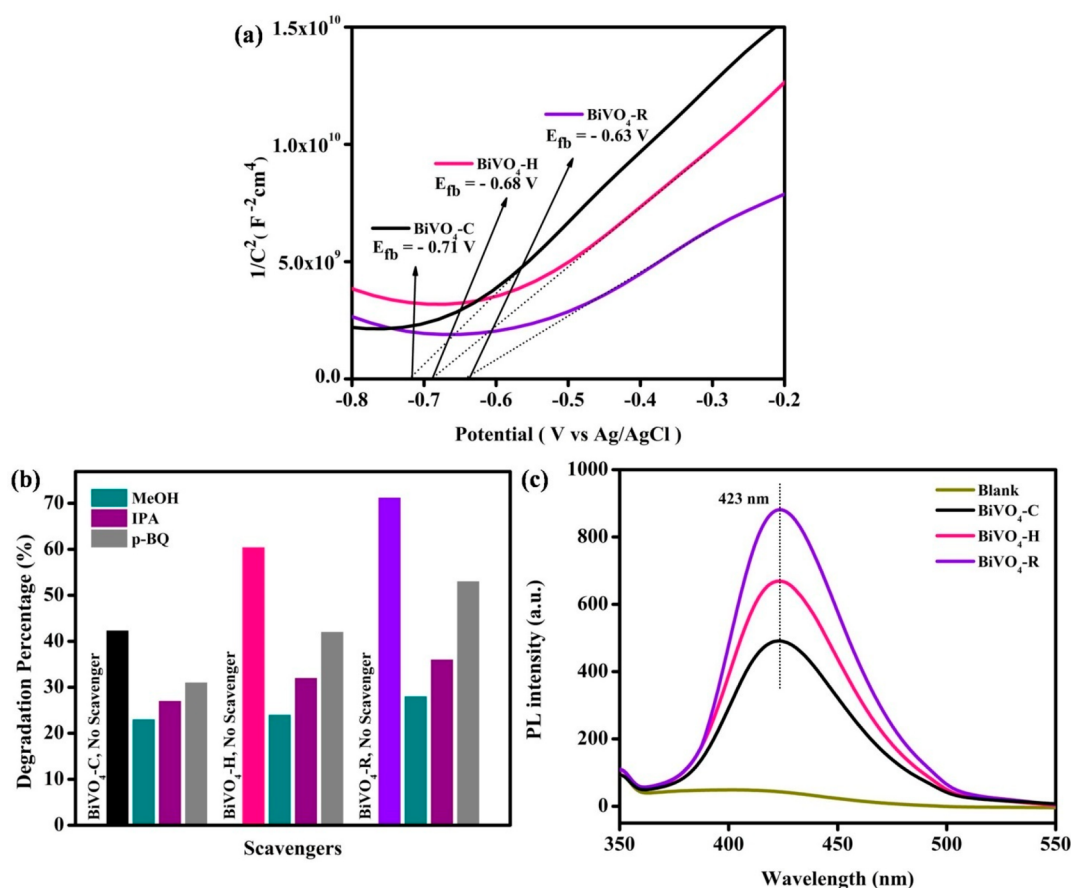
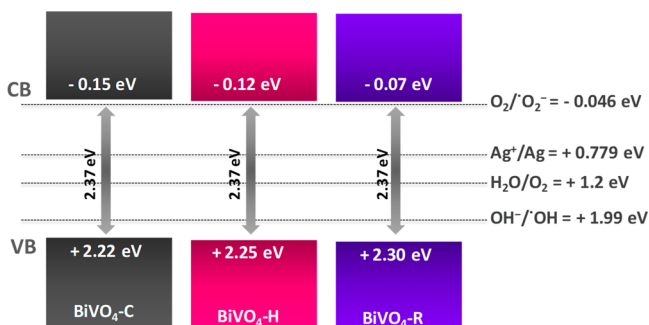


Figure 11. (a) Mott–Schottky, (b) active radical measurement, and (c) TA plots of fabricated monoclinic scheelite BiVO₄ samples.

variations in band edge potential (CB and VB) values of fabricated BiVO₄ materials are due to diverse facet exposure factors, as the surface atomic coordinations and arrangements of different exposed facets are different.⁵⁹ The evaluated probable near band edge structures of the fabricated samples with relative potentials of various components accountable for photocatalytic redox reactions are shown in Scheme 4.

Scheme 4. Schematic Illustration of near Possible Band Structures of Fabricated Monoclinic Scheelite BiVO₄ Samples



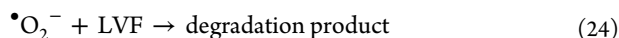
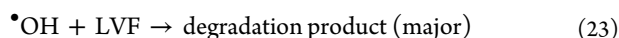
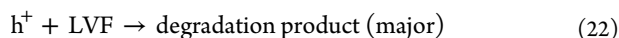
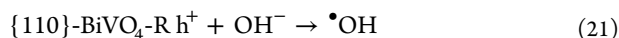
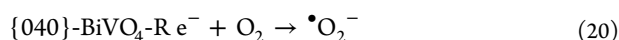
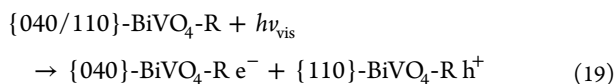
Moreover, the effective association of various active radicals and thus possible inherent mechanistic pathways involved in the detoxification process of levofloxacin were elucidated via sequential scavenging experiments. In the present investigation, h⁺, OH[•], and O₂⁻ active species are well trapped via the introduction of methanol (MeOH), isopropyl alcohol (IPA),

and *p*-benzoquinone (*p*-BQ) as scavenging agents, respectively (Figure 11b).^{11,12,22} As was observed, all three BiVO₄ samples with hierarchical morphologies demonstrate analogous outcomes. The detoxification was strappingly (strongly) hindered upon scavenging h⁺ and grounds (results) a lowering of detoxification to 23, 24, and 28% associated with BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively. Moreover, scavenging OH[•] radicals significantly decreases the degradation to 27, 32, and 36% associated with BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively. However, O₂⁻ makes a relatively lower contribution toward the degradation process and this is well-observed from a comparatively poorer decrease in degradation to 31, 42, and 53% for BiVO₄-C, BiVO₄-H, and BiVO₄-R, respectively. The results illustrate the involvement of each of the active radicals in the degradation process of LVF with a crucial contribution of h⁺ and OH[•] radicals. Moreover, a significant association of OH[•] radical was further validated from the considerable increase in PL intensity owing to the transformation of terephthalic acid (TA, 423 nm) to fluorescent hydroxyterephthalic acid (HTA) by the excess OH[•] radicals produced.⁶⁰ The results follow the trend BiVO₄-C < BiVO₄-H < BiVO₄-R, suggesting enhanced exciton pair separation over the BiVO₄-R sample (Figure 11c).⁶¹ The results were in good agreement with the band configurations of fabricated samples and previous reports (Scheme 4).^{34–36,61}

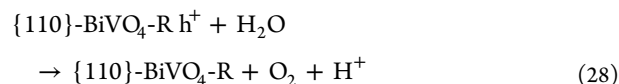
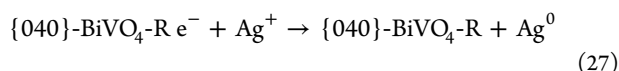
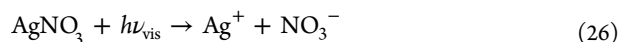
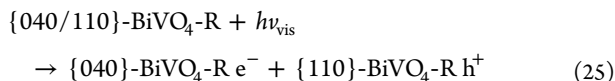
In this context, a credible mechanistic pathway was projected to better explain the photoredox phenomenon over BiVO₄-R. The inherent {040/110} facet isotype heterojunction promotes the augmented accumulation of photoexcited electrons over {040}-BiVO₄-R with a lower conduction band

potential and photogenerated holes over {110}-BiVO₄-R with a lower valence band potential, respectively.^{29,33} Meticulously, the holes over the BiVO₄-R {110} facet actively oxidize OH[−] to produce an [•]OH radical (OH[−]/[•]OH = +1.99 eV), while electrons over the BiVO₄-R {040} facet react with dissolved O₂ to produce the [•]O₂[−] radical (O₂/[•]O₂[−] = −0.046 eV). The h⁺, [•]OH, and [•]O₂[−] radical subsequently instigate the detoxification of levofloxacin (Scheme 3c-I).^{34,35} On the other hand, the excess electrons over the BiVO₄-R {040} facet with high reduction potential actively reduce Ag⁺ to form metallic Ag (Ag⁺/Ag⁰ = +0.779 eV). Consequently, the active holes over the BiVO₄-R {110} facet rapidly oxidize H₂O to O₂ (H₂O/O₂ = +1.2 eV) (Scheme 3c-II).^{11,12}

(I) Possible levofloxacin degradation pathway:



(II) Possible water oxidation pathway:



4. CONCLUSION

In summary, highly crystalline hierarchical microstructures of monoclinic scheelite BiVO₄ were fabricated via varied synthetic courses and were well established by analytical characterizations. An Annona reticulate-shaped BiVO₄ was obtained by a calcination (BiVO₄-C) treatment, while truncated bipyramidal morphologies with inbuilt {040/110} facet isotype heterojunctions were achieved by hydrothermal (BiVO₄-H) and reflux (BiVO₄-R) treatments. The photocatalytic redox efficiencies of the fabricated microstructures of BiVO₄ were well investigated over levofloxacin detoxification and the water oxidation reaction and actively compared. Superior exciton separation promoted enhanced photoactivity over the BiVO₄-R sample was exposed. The enhanced photoexciton anti-recombination phenomenon was attributed to (i) well-exposed {040} reduction facet and {110} oxidation facet promoted facet isotype heterojunction formation and (ii) a high exposure percentage of {040} reduction facets of BiVO₄-R. The effect of physicochemical features, photoelectrochemical properties, {040/110} facet isotype heterojunctions, and relative exposure factors of oxidation–reduction facets toward the superior

photocatalytic redox activity of BiVO₄ was well established. The findings are strongly promising and captivating toward the development of morphology and relative reduction–oxidation facet exposure factor selective fabrication of facet isotype heterojunction based materials for superior visible-active semiconductor photocatalysis.

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Notes

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